

Restoring the Density-Gradient Expansion for Exchange in Solids and Surfaces

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Popular modern generalized gradient approximations are biased toward the description of free-atom energies. Restoration of the first-principles gradient expansion for exchange over a wide range of density gradients eliminates this bias. We introduce a revised Perdew-Burke-Ernzerhof generalized gradient approximation that improves equilibrium properties of densely packed solids and their surfaces.

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Ground-state Kohn-Sham density functional theory [1] has been hugely successful for electronic structure calculations of solids and molecules. It relies upon good approximations to the exchange-correlation (xc) energy as a functional of the electronic spin densities [2]. Over four decades, increasingly sophisticated approximations have been developed [3]. The most commonly used in solid-state calculations today is the Perdew-Burke-Ernzerhof (PBE) version [4] of the generalized gradient approximation (GGA), employing both the density and its gradient at each point in space. Popular GGAs represent a well-tempered balance between computational efficiency, numerical accuracy, and reliability, but PBE also juggles the demands of quantum chemistry and solid-state physics [5].

While PBE represented a high point of nonempirical functional development 11 years ago, much has since been learned about its limitations. PBE reduces the chronic overbinding of the local spin density approximation (LSDA) [1], but, while LSDA often slightly underestimates equilibrium lattice constants by about 1%, PBE usually overestimates them by about the same amount. Other equilibrium properties, such as bulk moduli, phonon frequencies, magnetism, and ferroelectricity, are sensitive to the lattice constant, and so are also overcorrected by PBE [6]. Surface energies are too low in LSDA, but are made lower still by PBE [7].

However, attempts to construct a better GGA face a Procrustean dilemma [8]: Those with an enhanced gradient dependence [9,10] improve atomization and total energies, but worsen bond lengths, while more recent suggestions of a GGA for solids [6,11–14] (like the very first GGA [15]) have a reduced gradient dependence and typically do improve lattice parameters and/or surface energies, but have been criticized for worsening total and atomization energies [13]. More advanced functionals have been constructed. For example, meta-GGAs, using also the orbital kinetic-energy densities, provide greater accuracy over a wider range of systems and properties [3]. But current

meta-GGAs do not improve lattice constants as dramatically as surface energies, and meta-GGAs are not yet available in all solid-state codes.

In the present Letter, we explain the origin of this dilemma and show that *no* GGA can do both: Accurate atomic exchange energies require violating the gradient expansion for slowly-varying densities, which is valid for solids and their surfaces. At the GGA level, one must choose. A pragmatic approach to lattice properties is therefore to use a modified functional especially for solids which, unlike previous suggestions, recovers the gradient expansion for exchange over a wide range of density gradients. This by itself yields much better lattice constants than PBE (see Ref. [16]), and excellent jellium surface exchange energies, because such energies are dominated by moderate density gradients. Numerical studies [2] suggest that PBEsol will also yield highly accurate bulk exchange energies of metals within the pseudopotential approximation. These observations suggest an improved starting point for more advanced functional construction.

Restoration of the gradient expansion for exchange requires a complementary alteration for correlation. We fit the jellium surface xc energy, as done by Armiento and Mattsson [12]. This condition is insufficient to yield accurate lattice constants, as experience with the Tao-Perdew-Staroverov-Scuseria (TPSS) meta-GGA shows [3]. But, as explained below, it is relevant to lattice constants, because of its bulklike nature. Being accurate for both surface exchange and xc energies, PBEsol minimizes reliance on error cancellation between x and c , while Ref. [12] does not. By using the PBE form but simply altering two parameters, we retain all other exact conditions that make PBE so reliable.

The GGA form for the exchange energy is simply

$$E_x^{\text{GGA}}[n] = \int d^3r e_x^{\text{unif}}(n(\mathbf{r})) F_x(s(\mathbf{r})), \quad (1)$$

where $n(\mathbf{r})$ is the electronic density, $e_x^{\text{unif}}(n)$ is the exchange

energy density of a uniform electron gas ($\sim n^{4/3}$), $s = |\nabla n|/(2k_F n)$ [with $k_F = (3\pi^2 n)^{1/3}$] is the dimensionless density gradient, and $F_x(s)$ is the enhancement factor for the given GGA [4]. Equation (1) is the spin-unpolarized form, from which the spin dependence can be deduced [4]. Any GGA that recovers the uniform gas limit has

$$F_x(s) = 1 + \mu s^2 + \dots \quad (s \rightarrow 0). \quad (2)$$

The gradient expansion [1] that is accurate for slowly-varying electron gases has [17] $\mu_{\text{GE}} = 10/81 \approx 0.1235$.

To begin, Ref. [18] showed that the exchange energies of neutral atoms are very well approximated by their asymptotic expansion for large Z , i.e., $E_x = -0.2208Z^{5/3} - 0.196Z + \dots$. The first term arises from LSDA, but the second arises in a GGA from the s^2 contribution to Eq. (2) and requires $\mu \approx 2\mu_{\text{GE}}$. Thus any GGA that is accurate for the exchange energies of free neutral atoms must have $\mu \approx 2\mu_{\text{GE}}$. PBE does, although its value of $\mu = 0.2195$ was found from a different nonempirical argument. So does B88, as it was fitted to the x energies of noble gas atoms [19]. Even PW91 does [20], as it reverts only at irrelevantly small s to μ_{GE} .

Thus, to attain accurate exchange energies of atoms (vital to dissociation energies in molecules and cohesive energies in solids), any GGA must strongly violate the gradient expansion for slowly varying densities [18]. But most of thermochemistry occurs without free atoms, and is not much worse in LSDA than in PBE (e.g., [21]). Moreover, for the evaluation of exchange, the densities of real solids and their surfaces are often almost slowly varying over space. Restoring the gradient expansion should improve their description (but worsen atomization energies). The GGA is a limited form, and *cannot* satisfy both conditions. Equation (2) suggests a necessary condition for convergence of the second-order gradient expansion for exchange: $s \leq 1$. Since $s \leq 1$ for valence electrons in densely packed solids (or $s \leq 2$ in core-valence regions of alkali atoms), and since the reduced Laplacian of the density is also ≤ 1 , the gradient expansion is important for exchange in solids. We choose μ_{GE} for PBEsol.

Now, for a GGA correlation functional that recovers the uniform gas limit, the gradient expansion is

$$E_c[n] = \int d^3 r n(\mathbf{r}) \{ \epsilon_c^{\text{unif}}(n(\mathbf{r})) + \beta t^2(\mathbf{r}) + \dots \} \quad (3)$$

where $\epsilon_c^{\text{unif}}(n)$ is the correlation energy per particle of the uniform gas, β is a coefficient, and $t = |\nabla n|/\{2k_{\text{TF}}n\}$ is the appropriate reduced density gradient for correlation (fixed by the Thomas-Fermi screening wave vector $k_{\text{TF}} = \sqrt{4k_F/\pi}$, not k_F). For slowly varying high densities [22], $\beta_{\text{GE}} = 0.0667$. Unlike exchange, the second-order term in the gradient expansion for correlation cannot be small compared to the local term everywhere even for valence electrons in solids: βt^2 can be large compared to $|\epsilon_c^{\text{unif}}|$ (as $\beta_{\text{GE}} t^2 = 0.1 s^2/r_s$). The gradient expansion can be relevant

to real systems (especially solids) for exchange, but much less so for correlation.

Also relevant to solids is $f_{xc}(q)$ for the response of the uniform gas to a weak potential $\lambda \cos(\mathbf{q} \cdot \mathbf{r})$. For valence-electron densities, the exact $f_{xc}(q)$ is almost independent of q , up to $2k_F$ [23]. Thus LSDA, which produces a constant (the value at $q = 0$), yields an accurate approximation for $q \leq 2k_F$. But any GGA with a nonzero xc contribution to second order in ∇n produces a term quadratic in q . Since we are interested in weakly-varying valence-electron densities in densely packed solids, we wish to retain this excellent feature of LSDA. If

$$\mu = \pi^2 \beta / 3, \quad (4)$$

there is complete cancellation between beyond-LSDA x and c contributions, restoring LSDA response.

In PBE, the gradient expansion for correlation is respected, i.e., $\beta = \beta_{\text{GE}}$, and $\mu \approx 2\mu_{\text{GE}}$ satisfies Eq. (4). This choice agrees well with the PW91 exchange functional, and with its hole construction [20] (where sharp real-space cutoffs are most appropriate to atoms), and yields highly accurate exchange energies of atoms. But we have already argued that $\mu \approx 2\mu_{\text{GE}}$ is harmful for many condensed matter applications. Once we choose μ_{GE} for exchange, we cannot recover simultaneously the gradient expansion for correlation and the linear response of LSDA for a uniform density. Exact satisfaction of Eq. (4) would yield $\beta = 0.0375$, but a compromise value will satisfy another, more relevant constraint for solid-state applications.

For correlation, large neutral jellium clusters are our paradigm, for which $E_{xc} \rightarrow e_{xc}^{\text{unif}} V + e_{xc}^{\text{surf}} A + \dots$ as the radius grows, where e_{xc}^{surf} is the jellium surface xc energy, V the volume of the cluster, and A its area. A GGA that recovers e_{xc}^{surf} will be correct in leading- and next-order for neutral jellium clusters as $N \rightarrow \infty$, in a similar way to popular exchange GGAs for neutral atoms. Moreover, the surface energy is dominated by xc contributions and e_{xc}^{surf} is a bulklike property, arising mainly (103% at $r_s = 2$) from a moderately-varying-density region (with $s \leq 1.9$) inside the classical turning plane.

We check that this condition is compatible with the restoration of the gradient expansion for exchange. Because jellium clusters have a uniform bulk density and because most of the surface energy comes from within, the gradient expansion should be accurate. We find, at bulk density $r_s = 3$, the errors of the surface exchange energy are: LSDA 27%, PBE -11%, and PBEsol 2.7%.

We fit e_{xc}^{surf} [12] to determine our correlation functional. But the jellium e_{xc}^{surf} is not known exactly. Figure 1 shows surface energy enhancements relative to LSDA. The likely “range of the possible” for e_{xc}^{surf} extends from TPSS meta-GGA [3,7] or RPA+ [24] at the low end of what is possible (in agreement with the most recent quantum Monte Carlo calculations [25]) to the RPA-like Pitarke-Perdew (PP)

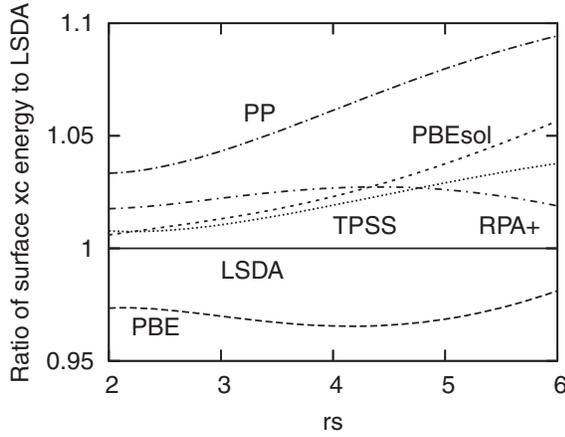


FIG. 1. Ratio of calculated surface exchange-correlation energy to that of LSDA as a function of r_s for various approximations.

[26] value at the high end. For a history of this subject, and another sophisticated estimate, see Ref. [27]. TPSS may provide the best target for a GGA. We choose $\beta = 0.046$ and $\mu = \mu_{GE}$ (within the PBE form) for PBEsol, to best fit the TPSS results. PBEsol should improve most surface energies over LSDA, whereas PBE worsens them.

Thus we have violated Eq. (4) in favor of good surface energies. But our value for β is considerably closer to that of the linear response requirement (0.0375) than that demanded by complete restoration of the gradient expansion (0.0667).

PBEsol becomes exact for solids under intense compression, where real solids and their surfaces become truly slowly varying, and exchange dominates over correlation [18]. In Fig. 2, we plot the enhancement factors of PBE and PBEsol. For a spin-unpolarized ($\zeta = 0$) density $n = 3/(4\pi r_s^3)$, we define $F_{xc}(r_s, s)$ by

$$E_{xc}^{GGA}[n] = \int d^3r e_x^{\text{unif}}(n(\mathbf{r})) F_{xc}(r_s(\mathbf{r}), s(\mathbf{r})). \quad (5)$$

The high-density ($r_s \rightarrow 0$) limit is $F_x(s)$ of Eq. (1). The nonlocality or s dependence of GGA exchange is diminished from PBE to PBEsol, making the latter somewhat closer to LSDA. Over the whole range $s \lesssim 1$, the PBEsol F_x is close to $1 + \mu_{GE}s^2$. The range $0 \lesssim s \lesssim 3$ is energetically important for most properties of most real systems, while $0 \lesssim s \lesssim 1$ and $1 < r_s < 10$ are the ranges for valence-electron regions in many densely packed solids.

To test our functional, we employ a test set of 18 solids from Ref. [28]. These come in four groups: simple metals (Li, Na, K, Al), semiconductors (C, Si, SiC, Ge, GaAs), ionic solids (NaF, NaCl, LiCl, LiF, MgO), and transition metals (Cu, Rh, Pd, Ag). The set is not claimed to be representative, but was chosen for the availability of basis functions and anharmonic corrections [28]. Our calculations use the Gaussian orbital periodic code of Ref. [28],

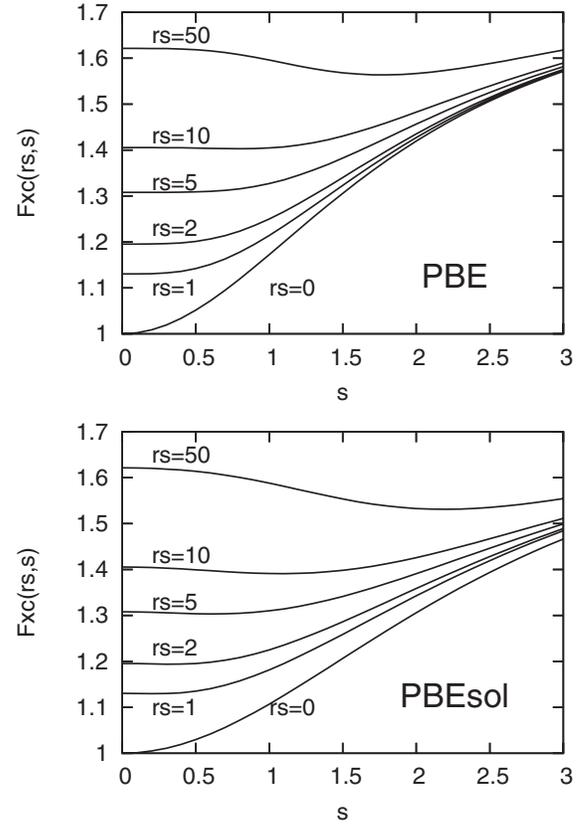


FIG. 2. Enhancement factors of PBE and PBEsol, for spin-unpolarized systems, as a function of reduced density gradient, for various values of r_s .

with basis sets of the same or higher quality. In Table I, we list both mean and mean absolute errors for lattice constants. The systematic PBE overestimate is close to the systematic LSDA underestimate, as shown by the total

TABLE I. Errors in equilibrium lattice constants (in $\text{\AA} \times 10^{-2}$) on our data set of 18 solids, relative to experiment with estimates of the zero-point anharmonic expansion removed [28].

Class	LSDA	PBE	TPSS	PBEsol
Mean error				
4 simple metals	-9.0	2.9	5.3	-0.3
5 semiconductors	-1.1	7.9	6.2	3.0
5 ionic solids	-8.4	8.5	6.8	2.0
4 transition metals	-4.0	6.4	2.5	0.0
Total	-5.5	6.6	5.4	1.3
Mean absolute error				
4 simple metals	9.0	3.4	5.3	2.3
5 semiconductors	1.3	7.9	6.2	3.0
5 ionic solids	8.4	8.5	6.8	2.7
4 transition metals	4.0	6.4	2.7	1.9
Total	5.6	6.7	5.4	2.5

TABLE II. Errors in atomization energies (eV) for the AE6 set of molecules, using the 6-311 + $G(3df, 2p)$ basis set.

Error	LSDA	PBE	TPSS	PBEsol
Mean error	3.35	0.54	0.18	1.56
Mean abs. error	3.35	0.67	0.26	1.56

mean absolute errors, and TPSS cures this very little. On the other hand, PBEsol greatly reduces this overestimate, by a factor of almost 4, except for semiconductors, where LSDA is unsurpassed.

PBEsol is not expected to give good atomization energies. In Table II, we give the errors on the AE6 data set of molecules. These 6 molecules [SiH_4 , S_2 , SiO , C_3H_4 (propyne), $\text{C}_2\text{H}_2\text{O}_2$ (glyoxal), and C_4H_8 (cyclobutane)] were chosen [29] to be representative, i.e., to reproduce the errors of much larger data sets. As is clear, and expected, PBEsol is much less accurate than PBE, only about halving the error of LSDA. This can be related to PBEsol's worsened total energies of atoms.

We have demonstrated the relevance of the second-order gradient coefficient for the exchange energy of a slowly varying density to the bulk and surface properties of solids. The TPSS meta-GGA [3], which incorporates this coefficient, gets good surface energies, but its lattice constants are only marginally better than those of PBE on which it builds, whereas PBEsol is significantly better. This suggests that an improved meta-GGA needs to recover the gradient expansion for exchange over a wider range of density distributions $n(\mathbf{r})$ than TPSS does.

Previous attempts to improve on PBE within the GGA form have retained the PBE gradient coefficients μ and β for small s , but altered the behavior at large s [6,9,10], or have zeroed out μ [11,12], and are thus fundamentally different from PBEsol. The AM05 [12] functional performs very similarly to PBEsol for the solids studied here, but AM05 follows the proposal of Vitos *et al.* [11] to fit the conventional exchange energy density of an Airy gas. For $r_s = 2$ to 6, AM05 exchange has errors of 12% to 60% for e_x^{surf} , compared to 1.6% to 4.1% for PBEsol. Numerical comparisons and details are available [16].

We have identified the simple exchange-correlation physics underlying many properties of many solids, and shown how it differs from that for atoms. We recommend PBEsol for the applications discussed here. Any existing code that implements PBE can be instantly modified to try PBEsol, by simply replacing the values of μ and β . Modified PBE subroutines are available from [30].

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Erratum: Restoring the Density-Gradient Expansion for Exchange in Solids and Surfaces [Phys. Rev. Lett. 100, 136406 (2008)]

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Table I of this Letter and Table SI of its Supplementary Information require small corrections to the corrected experimental values of the lattice constants for the nonmetals.

The errors of the density-functional lattice constants in Table I were deviations from corrected experimental values, where the corrections removed the finite-temperature and zero-point anharmonic expansion (ZPAE) effects. The ZPAE corrections were taken from Ref. [28] and based on Eq. (18) of Ref. [28], i.e., on Appendix A of A. B. Alchagirov *et al.*, Phys. Rev. B **63**, 224115 (2001), in which ε is the energy per atom and v_0 is the volume per atom. But Ref. 28 inadvertently used the volume per unit cell, which is twice the volume per atom for the semiconductors and ionic crystals. Thus, the ZPAE corrections for these solids in Ref. [28] and in our article were only half what they should have been. Parts of Table I (with errors in units of 0.01 Å) of our article should be corrected as follows:

Class	LSDA	PBE	TPSS	PBEsol
Mean error				
5 semiconductors	-0.3	8.6	7.0	3.7
5 ionic solids	-6.9	10.1	8.3	3.5
Total	-4.9	7.3	6.0	1.9
Mean absolute error				
5 semiconductors	0.9	8.6	7.0	3.7
5 ionic solids	6.9	10.1	8.3	3.5
Total	5.1	7.4	6.0	2.9

The corrected experimental lattice constants for the 4 simple metals and 4 transition metals in Table I are unaffected. No conclusion is changed by this erratum, although the improvement of PBEsol relative to LSDA is slightly reduced. Some lattice constants (in Å) from the “Expt.-ZPAE” column of Table S1 of the Supplementary Information (EPAPS PRLTAO-100-016814) should be corrected as follows: 3.544 (C), 5.416 (Si), 4.340 (SiC), 5.640 (Ge), 5.638 (GaAs), 5.566 (NaCl), 4.579 (NaF), 5.074 (LiCl), 3.964 (LiF), and 4.188 (MgO).

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